Feasibility of luminescent multilayer sol-gel thermal barrier coating manufacturing for future applications in through-thickness temperature gradient sensing

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A B S T R A C T

This paper investigates the feasibility of manufacturing sol-gel multilayer thermal barrier coatings (TBC) functionalized with different lanthanide ions Ln3+ having distinct photo-luminescence emission wavelengths (Ln = Sm, Eu, Dy, Er, Tm) for future applications in temperature gradient sensing. Ln3+ doped 9.75 mol% yttria stabilized zirconia (YSZ) powders were produced to study the effect of activator concentration on luminescence intensity and host matrix crystal structure. Self-quenching was found to limit the maximum signal-to-noise ratio achievable with Sm3+, Dy3+, Er3+ and Tm3+ activators, which was not the case for Eu3+ in the 1–10 mol% range. The increase in activator was found to affect the crystal structure of YSZ. A solution was proposed that suppressed this effect while significantly increasing the luminescence intensity of all activators. Finally a TBC sensor prototype integrating Eu3+, Er3+ and Dy3+ doped layers distributed throughout the thickness was successfully deposited by a dip-coating sol-gel process and showed promising through-thickness luminescence sensing capabilities.

1. Introduction

The progressive use and improvement of thermal barrier coatings (TBCs) to protect critical components from heat damage in gas turbines have allowed significant increase in turbine inlet temperature over the last 30 years [1,2]. Today’s jet engines’ efficiency and durability therefore greatly depend on TBC’s reliability and performance. The degradation during service of the current standard 6–8 wt% yttria stabilized zirconia (YSZ) based TBCs is mainly caused by the initiation and the propagation of microcracks at the interface between either the bond coat and the thermally grown oxide (TGO) or the TGO and the ceramic overcoat resulting from the thermal expansion coefficient mismatch between coatings and substrates [3,4]. This sub-surface mode of failure makes an early detection particularly difficult. The development of reliable predictive models for YSZ overcoat spallation is indeed limited by the difficulty of accessing the effective interface temperature that governs this process through conventional means without compromising the integrity of the coating. This results in strongly conservative margins being imposed to allow safe operation of jet engine turbine components.

From this perspective there has been a growing interest in the application of phosphor thermometry methods for the diagnostic of TBCs as the partial transparency of YSZ in the visible range of the spectrum allows to collect local information conveyed by the luminescence emissions from optically excited luminescent layers integrated throughout the depth of the TBC [5,6]. This functionalization can be obtained by the introduction of optically active components such as trivalent lanthanide ions directly into the crystal structure of YSZ, thus without any detrimental alterations of the coating properties [6–8]. Such “sensor TBCs” has shown high potential for measuring substrate/TBC interface temperature [9,10] or investigate local interface delamination [11,12] by non intrusive optical sensing.

Reported here is a feasibility study for the design and manufacturing of multilayer functionalized TBCs, deposited by a dip-coating process from sol-gel precursors, alternative to the standard electron beam physical vapor deposition (EB-PVD) and atmospheric plasma spraying (APS) methods used for YSZ coatings, for future applications in through-thickness measuring of temperature gradients. This cost effective process allows one to deposit 9.75 mol% porous YSZ TBCs presenting the metastable ‘t’ phase with performance under cyclic oxidation comparable to that of EB-PVD coatings [13,14]. The easy composition control offered by the technique has already generated interest for manufacturing some high purity YSZ:Er and YSZ:Sm phosphor coatings and powders [15]. Therefore the effect of additions of several dopants including Sm, Eu, Dy, Er and Tm in 9,75 mol% YSZ was investigated, and a solution is proposed for the optimization of YSZ luminescence properties. A prototype of multilayer TBC with three luminescent layers with distinct
luminescence emissions embedded throughout the thickness, respectively doped with Eu, Er and Dy, was deposited by dip-coating and characterized to assess the feasibility of the process.

2. Thermal barrier coating sensors

The health monitoring issues brought about by the critical role played by TBCs in the durability of jet engines led to the development of the concept of multifunction TBCs, that combine thermal insulating and optical sensing capabilities. The latter are based on the semi-transparency of zirconia in the visible range and its ability to accept small amounts of trivalent lanthanide ions that substitute for Zr\(^{4+}\) cations in its lattice and act like luminescent activators under appropriate UV or visible illumination (Fig. 1), thus providing information about their local environment. First introduced by Amano et al [16] in 1988, multilayer TBCs alternating various luminescent and undoped layers allow one to evaluate the progression of TBC erosion from the appearance/delamination areas at the interface region in APS-TBCs integrating a 7–10μm thick luminescent sublayer. Finally the application of phosphor thermometry methods [19,20], mainly based on the dependence of luminescence intensity and decay time on temperature, allows non-intrusive optical sensing of buried luminescent layers to determine through-thickness temperature. Application to bilayer systems such as YSZ: Dy/YSZ [9,21] and YSZ: Eu/YSZ [10] has thus allowed measurements of substrate/YSZ interface temperature. The existence of several dopants with distinct luminescence emission peaks suggests that more complex layered TBC architectures, also referred by Gentleman and al. [6] as “rainbow sensors”, could be powerful tools to measure the complete temperature gradient or heat flux across TBCs. Fig. 1 presents the luminescence spectra of the various luminescent activators in 9.75-YSZ selected for this work on the basis of their luminescence properties and performance as thermographic phosphors either in YSZ or other matrices. Several doublets and triplets of compatible activators such as Tm-Er-Eu or Dy-Er-Eu are thus indentified as valid candidates to build multilayer TBC sensors for through thickness health and temperature monitoring. It should be born in mind that although Dy\(^{3+}\) and Tm\(^{3+}\) appear to have distinct emissions, thermalization of the \(\bar{A}_{1/2}\) level of Dy\(^{3+}\) at high temperature will generate an additional peak at 458 nm potentially overlapping the peaks of Tm\(^{3+}\) [22], limiting to three the number of functionalized luminescent layers that could be used for this range of activators. Of all dopants presented in Fig. 1, Eu\(^{3+}\), Er\(^{3+}\) and Dy\(^{3+}\) exhibit the brightest luminescence in YSZ in addition to having distinct emission peaks in the red, green and blue region of the visible spectrum respectively, that therefore make them the best candidates for the design and the manufacturing of a multilayer TBC sensor prototype.

3. Experimentation

3.1. Synthesis of YSZ phosphors

YSZ phosphor powders (9.75 mol\% YO\(_1\)\(_2\)\(_3\)) with contents of Ln\(_{3+}\) (Ln = Eu, Er, Dy, Sm or Tm) between 0.15 and 10 at% were synthesized using the sol-gel route [23]. The sols were prepared by mixing zirconium (IV) propoxide (Zr(OPr)\(_4\)) (Sigma Aldrich) and appropriate contents of yttrium (III) and (III) nitrates (Sigma Aldrich) as precursors in a solution of 1-propanol (Sigma Aldrich) and ultrapure water. Acetylacetonate (AcAc) (Sigma Aldrich) was used as a complexing agent to control the kinetics of hydrolysis of the zirconium alkoxide [24]. The volume rates of [AcAc:Zr(OPr)\(_4\)] and [H\(_2\)O/Zr(OPr)\(_4\)] were kept constant at 0.8 and 9.5 respectively. More details about the preparation of the mixtures from the reagents can be found in reference [23]. After 30 min of mechanical stirring the solutions were then held for one night (~16 h) at 50 °C to accelerate the hydrolysis and condensation of the sols into bright monolith gels free of precipitates. The solvent was then evaporated in a drying oven at 70 °C. The resulting dried gels, called xerogels, were subsequently calcined for 2 h at 800 °C to remove the remaining traces of solvents and then manually ground to powders for 4 min. The products obtained at this stage are white powders consisting in 2–50 μm particles of partially crystallized YSZ. An additional heat treatment of the powders at 1100 °C for 2 h ensured the full crystallization of the YSZ into metastable quadratic phase \(\gamma'\) or cubic phase \(\gamma\) depending on their composition.

3.2. Deposition of TBC sensors

Multilayer TBC sensor prototypes containing luminescent layers of composition Ln\(_{0.0575-\text{Zr}_{1.951-2x}}\text{YO}_{1.092-x}\text{O}_{1.951-2x}\) (Lnm = Eu, Er or Dy) with distinct emissions wavelengths were shaped by a dip-coating process previously developed [14,15].

First YSZ xerogel powders were synthesized from a sol-gel protocol adapted from Lecompte et al. [25]. Rare earth doped YSZ gels were prepared as described in part 3.1 but were dried above the supercritical point of 1-propanol (T\(_c\) = 261 °C; P\(_c\) = 5.1 MPa) in a stainless steel autoclave (Paar Instrument 4621) instead of being dried in free air. Temperature was kept constant at 270 °C and pressure at 9 MPa for 1 h, afterwards the solvent was slowly released in isothermal conditions. The resulting brittle monolithic aerogels whose structure is similar to that of the wet gels were calcined for 2 h at 700 °C and ball milled at 250 rpm for 1 h in an agate mortar to obtain highly porous powders having morphological and grain size characteristics more suitable for dispersion and deposition by dip-coating than xerogel powders obtained through conventional drying at atmospheric pressure [23,26].

The substrates used for the depositions were AM1 single crystal nickel based superalloy coupons coated with Ni(PlAu) bondcoat, previously sand blasted with 10 μm corindon particles and precoxidized for 2 h at 950 °C under a O\(_2\) pressure of 0.05 Pa. Slurries were prepared by mixing 20 wt% of YSZ sol with YSZ aerogel powder previously dispersed in 1-propanol with polyvinylpyrrolidone 3500 (PVP) (Acros Organics) dispersing agent. The depositions were carried out at room temperature by dipping the substrates in the slurry with a controlled withdrawal rate of 250 mm/min to ensure the deposition of a -10–15 μm thick layer at each dip procedure. The deposited films were then dried for 5 min at 50 °C before implementing the next dip-coating.

Coatings of final thickness in the range of 100–150 microns were obtained after 13 successive dips. Multilayer prototypes alternating...
luminescent and unmodified YSZ layers such as shown in Fig. 2 were built by successive dipping into slurries with different compositions. Subsequently, samples were sintered for 2 h at 1100 °C. The heating and cooling rates were set at 30 °C/h to minimize thermal stresses caused by the mismatch between the thermal expansion coefficients of the coating and the substrate.

3.3. Characterization

The luminescence spectra of both powders and coatings were recorded at room temperature over the 400–700 nm range with an Ocean Optics USB2000 spectrometer equipped with a 1000 µm optical fiber. A Philips TL 6 W BLB mercury vapor lamp (main UV emission peak at 365 nm) and a 1.1 W continuous wave diode pumped solid state green laser operating at 532 nm (CNI laser) were used as excitation sources of the phosphors. A color CCD camera (AVT PIKE F-145C) was used to record images of the luminescence intensity emitted by the embedded luminescent sublayers. A narrow band interference filter centered at 636 nm (FWHM ~ 10 nm) was also used to filter the luminescence signal from the YSZ:Eu³⁺ layer.

Structural analysis of YSZ powders and coatings were performed by X-rays diffraction (XRD) measurements. XRD patterns were collected by scanning the angular range from 25° to 100° with a PANalytical X’Pert diffractometer equipped with a X’Celerator linear detector using Cu Kα (λ(Cu Kα1) = 1.5406 Å; λ(Cu Kα2) = 1.5445 Å) as the X-ray source. The operating voltage and currents were respectively 45 kV and 40 mA, with a step size of 0.08°/2θ and a step time of 0.25 s. The determination of crystal parameters was performed by Rietveld refinement [27,28] of the DRX patterns with the commercial software PANalytical X’Pert HighScore Plus using the P4/mmc and Fm3m space groups for the quadratic and cubic phases respectively. Scanning electron microscope (SEM) observations of the powders and coatings cross sections were achieved using a FEI Nova NanoSEM 450 field emission gun scanning electron microscope (FEG-SEM) operating at an accelerating voltage of 10 kV. The position and thickness of the different layers were determined from energy-dispersive X-ray (EDX) measurements performed on multilayer coating cross sections with the same SEM.

4. Results and discussion

4.1. Effects on luminescence intensity

Fig. 1 gives a representative example of the luminescence spectra from the different phosphors under continuous long-wave UV illumination of a mercury vapor lamp (365 nm). As the concentration in trivalent lanthanide ion activators was modified the only change observed in the luminescence spectra, not shown here, was the evolution of the intensities of the main luminescence peaks. This evolution is plotted in Fig. 3. The intensities of the main peaks were normalized by the corresponding maximum intensity in each spectrum. 9.75-YSZ phosphors doped with Dy, Sm, Er and Tm all exhibit an optimum of the intensity of the luminescence peaks for concentrations between 1 and 1.63 at%, after which further increases in activator content lead to lower intensities as a result of an increase of radiationless energy transfers between close neighboring dopant ions, a process also referred to as concentration quenching or self-quenching. The plotted lines in Fig. 3 represent the fit of the experimental data points with the direct excitation, self-quenching model by Johnson-Williams [31,32] relating the luminescence efficiency η of the activator (assumed to be proportional to the observed relative intensity) and the activator concentration C by the expression

$$\eta = \frac{C(1-C)^2}{C + D(1-C)}.$$  \hspace{1cm} (1)

Z and α/α are two fitting parameters, the former standing for the number of lattice positions (interstitial or substitutional) in the vicinity of an activator ion such that if another activator occupies one of these positions no luminescence will occur, and the latter standing for the ratio of the capture cross sections of photons by the host lattice and the activators.

The values obtained typically between 17–30 and 0.003–0.007 for Z and α/α respectively are consistent with reported values in the literature for YSZ:Ln type phosphors [6]. Luminescence output of these phosphors can thus be optimized with relatively limited composition modifications of the host material. The spectra given as an example in Fig. 1 provide a comparison of the luminescence strength of the different dopants Sm³⁺, Dy³⁺, Er³⁺ and Tm³⁺ at their optimal concentration in 9.75 at% YSZ mentioned above. Interestingly one can notice that Er³⁺ appears to emit a significantly brighter intensity than Dy³⁺, Sm³⁺ and Tm³⁺ in particular that host matrix.

It can be noticed in Fig. 3d that conversely to the other activators no self-quenching of luminescence is observed for YSZ:Eu in the 1–10 at% range, for which the luminescence intensity of the spectral lines is observed to linearly increase by one order of magnitude in that range. This behavior can be explained by the difference in self-quenching
properties between Eu$^{3+}$ ions and the other dopants investigated. While the presence of intermediate states between the initial excited state and the ground state in the energy level distribution of Er$^{3+}$, Sm$^{3+}$, Dy$^{3+}$ and Tm$^{3+}$ favors self-quenching at relatively low concentrations by cross relaxation, the large gap in energy between the top $^{5}D_{0}$ emitting level from which the 591 nm and 606 nm are originated (~17 250 cm$^{-1}$) and the top of the $^{7}F_{4,0}$ ground multiplet (~5000 cm$^{-1}$) of Eu$^{3+}$ electronic structure makes concentration quenching by this multipolar transfer process considerably less probable [7,33–35]. The self-quenching of Eu$^{3+}$ rather occurs by non-resonant transfer interactions strongly dependent upon crystal structure [33] and is not expected to occur in a YSZ matrix at concentrations below 10 at% [36]. Therefore intensity emissions up to more than 4 times that of other activators can be achieved with a 10 mol% addition of Eu$^{3+}$. Since Eu$^{3+}$ can also stabilize the desirable metastable tetragonal phase τ of ZrO$_2$ in the same way as Y$^{3+}$ does [37–39] it potentially offers more flexibility for the design of strongly emitting YSZ phosphors, although it might be limited by the potential effect of high concentration on luminescence, known to lead to quicker and more complex luminescence lifetime decays [7,19]. This latter aspect is still currently under investigation for the range of phosphors presented here.

4.1.2. Effects on YSZ crystal structure

The crystal structure of the various phosphor powders doped with Sm$^{3+}$, Eu$^{3+}$, Dy$^{3+}$, Er$^{3+}$ and Tm$^{3+}$ was investigated by X-Ray diffraction (XRD). Fig. 4a shows the three diffraction patterns of both undoped and Er$^{3+}$ doped 9.75-YSZ xerogel powders in the [30°–100°] 2θ range. The diffraction peaks observed on the un-doped YSZ pattern and the calculated c/√2 of 1.0096 correspond to the metastable τ crystal structure expected for YSZ with this composition prepared by this sol-gel route [23,40]. The diffraction patterns of powders doped with

![Fig. 3. Evolution of the intensities of the main luminescence peaks normalized by the corresponding maximum intensity for each spectrum with the concentration of a) DyO$_{1.5}$, b) SmO$_{1.5}$, c)ErO$_{1.5}$, d) EuO$_{1.5}$ and e) TmO$_{1.5}$ in 9.7at% YSZ xerogel powders. Plotted lines correspond to the fit of the experimental data points with the Johnson-Williams model [31,32] (Eq. (1)).](image1)

![Fig. 4. Diffraction patterns for 9.7 mol% YSZ:Er powders with increasing content of ErO$_{1.5}$ a) across the complete angular range, b) limited to the 72–78° range.](image2)
1.47 and 3 mol% confirm that they also crystallize in the t’ structure. However some changes are noticeable for some of the high angle peaks. In particular it can be observed in Fig. 4.b that the angular distance between the (004)_t and (220)_t reflections of the t’ phase initially at 73.2° and 74.3° tends to decrease with increasing Er content. These two peaks end up almost totally convolved for 3 mol% of Er³⁺. The same observation is made for the doublets of peaks at 35°, 60° and 84° 2θ angles. These results indicate that the material tends towards a cubic structure [40,41], a tendency also observed for similar concentrations in the Sm, Eu and Tm doped YSZ powders.

The evolution of the c/a/√2 values calculated from the refinement of the spectra by the Rietveld method reported in Fig. 5 is consistent with these observations. For all dopants the c/a/√2 ratio decreases linearly from the initial value of 1.0096 to nearly one in the range 0.5–6 mol%, as the quadratic phase progressively turns into the undesirable cubic phase. Finally, a single cubic phase is obtained for powders with Eu³⁺ content above 6 mol%, and the trends observed in Fig. 5 suggest that a single cubic phase would be obtained with similar contents of Sm³⁺, Dy³⁺, Er³⁺ or Tm³⁺. This linear decrease of the c/a/√2 ratio from 1.01 to 1 is typically reported in the literature for ZrO₂ doped with more than 10 mol% of YO₁₅. The minimum amount to ensure complete stabilization of the cubic phase being around 15 to 16 mol% [23,40]. These values are close to the corresponding 15.75 at% overall dopant concentration (YO₁₅ + EuO₁₅) required to fully stabilize the cubic phase in the 9.75-YSZ:Eu³⁺ phosphor powders presented here. Therefore it is concluded that the significant impact on the tetragonality of the quadratic phase at the relatively low concentrations required for optimizing luminescence intensity (1–1.63 mol%) is rather an effect of the overall substitution of Zr⁴⁺ by trivalent ions having larger ionic radius than an effect of the specific nature of the dopants. The functionalization of 9.75-YSZ without significant alteration of the sought crystal structure for TBC applications is thus limited by the already high content of YO₁₅ necessary to stabilize the desirable metastable tetragonal phase t’. This is particularly limiting for YSZ:Eu³⁺ that was shown to exhibit large increases in luminescence intensity in the range of 1–10 mol%, for which compromises appear to be necessary for a good balance between luminescent and structural properties.

4.2. Optimization of YSZ phosphors

In order to minimize the impact of rare earth additions in YSZ crystal structure, it was proposed to keep constant the overall dopant content in the lattice, i.e. to partially substitute the Y³⁺ stabilizing ions by luminescent markers. This solution was motivated by the fact that a minimum of 8 mol% YO₁₅ only is necessary for the synthesis of quadratic zirconia by the sol-gel route used in this work [40] and by the t’-ZrO₂ stabilizing ability—reported by various authors [37-39,42,43]—of some of these elements such as Eu³⁺ or Dy³⁺. The XRD patterns in the 2θ range 70–90° of 9.75-YSZ and 7.75-YSZ powders containing 2 mol% Eu³⁺ are compared to that of un-doped YSZ in Fig. 6. As seen previously in part 4.1.2, additions of 2 mol% of Eu to 9.75-YSZ cause a significant distortion of the unit cell, illustrated in Fig. 6 by the convolution of the (004)_t and (220)_t diffraction peaks at 74° as well as the low value of the calculated c/a/√2 ratio (1.0055). This effect is suppressed by the substitution of 2 mol% of Y³⁺ by the same amount of Eu³⁺, that thus preserves a t’ phase presenting a c/a/√2 ratio (1.0105) very close to that of the non-functionalized material (1.0102). Furthermore, further increasing the substitution rate to 4 mol% also appears to have very little additional effects on the crystal structure, suggesting the suitability of this solution for the fabrication of high luminescence efficiency YSZ:Eu phosphors. The crystal lattice parameters calculated for all the YSZ:Ln aerogel phosphor powders investigated are displayed in Table 1, and the comparison with that of un-doped 9.75YSZ powders confirms the relevance of Y³⁺ partial substitution for minimizing the impact of activator introduction on the microstructure (c/a/√2 ratios close to 1.01). Finally the luminescence spectra presented in Fig. 7 illustrate a quite interesting side effect of Y³⁺ substitution by Ln³⁺. Indeed lowering the Y³⁺ content by 2 mol% generates a 53% increase of the maximum peak intensity of the D₇₋→F₅ (591 nm) and D₆₋→F₂ (606 nm) transitions in YSZ:Eu 2 mol% aerogel powders. This effect appears to be dependent on Y³⁺ concentration as observations on 5.75-YSZ:Eu 4 mol% showed that the intensity corresponding to the above mentioned transitions increased by more than 60%. It was also observed that the emissions of all the other activators investigated (Table 1) are also enhanced by 8 to 30%, depending on the nature of the activator and the luminescence peak. At this stage it is still unclear whether this effect is related to the quenching of the luminescence by the Y³⁺ ions (or the associated oxygen vacancies), or is the result of changes in the optical properties of the host material in the emitting/ excitation range of the rare earth activators (transmission, absorption).

4.3. Characterization of a multilayer TBC sensor prototype

To investigate the feasibility of multilayer TBC sensor manufacturing from sol-gel precursors, 100 to 150 μm thick prototypes containing three thin luminene layers with distinct emissions wavelengths (respectively doped with Eu³⁺, Er³⁺ and Dy³⁺) distributed throughout the thickness were deposited on nickel superalloy coupons by dip-coating (Fig. 2). After sintering at 950 °C for 2 hours, smooth and uniform white coatings were obtained. XRD analysis confirmed that they all possess the desired metastable quadratic t’ structure with a c/a/√2 ratio of 1.0104. Fig. 8 shows SEM cross-section micrographs of a representative 100 μm prototype. The coating presents the typical microstructure of a sol-gel TBC consisting of a highly porous network of aerogel particles issued from the initial slurry bonded together by the crystalized sol (Fig. 8.b). The isotropic nature of the resulting porosity is believed to
provide a good balance between thermal insulation and thermomechanical strength [13, 14, 44].

The position and thickness of the different layers determined by EDX analysis are indicated in Fig. 8.a. There were no signs of adherence loss or crack initiation and propagation at interfaces between layers or between the substrate and the deposit, indicating the satisfactory sintering of the coating as a morphologically uniform single layer well adherent to the substrate. The main concern regarding the design of TBC sensor is the individual thickness and thickness uniformity of the luminescent layers. Since TBCs are subject to large thermal gradients, uniform sensing layer thicknesses of about 10 μm of below are to be preferred to avoid significant errors in the temperature readings [20]. There are in general little thickness variations within each layers, although the presence of some large particles (dimensions > 15–30 μm) can locally disturb the layers’ uniformity and density, either because of their large dimension or by contributing to the creation of large pores. However it should be noticed that the upper layers present a significantly larger thickness than the lower ones for the same deposition conditions, as the increase of the viscosity of the sol with time results in some noticeable film thickening with increasing number of dips. Nevertheless this effect can be easily controlled by adapting the number of dips for each layer or the withdrawal speed.

Fig. 9 presents the luminescence spectra of a coating under continuous UV illumination. The peaks corresponding to the YSZ:Dy top layer are clearly visible at 584–590 nm and 480–496 nm. However, the two doublets of peaks corresponding to the luminescent transitions in Er³⁺ at 516–524 nm and 544–561 nm are barely distinguished from the background, and no luminescence peaks from the YSZ:Eu layer are observed. Indeed the low power UV illumination source, that emits in a low transmission range of YSZ (365 nm) [5, 6], does not provide enough energy to the deepest layers to either excite the activators or generate a sufficiently strong luminescence output from the coating. Fig. 10 shows a filtered image of the TBC prototype under the illumination of a continuous high power laser operating at 532 nm, a wavelength that allows to stimulate both Er³⁺ and Eu³⁺ luminescence [6]. Although the 5D⁰ → 7 F₂ transition of Eu³⁺ at 636 nm, on which is centered the narrowband filter used, is 4 times less bright than the 5D⁰ → 7 F₂ emission line at 606 nm, the luminescence of the YSZ:Eu³⁺ layer at the interface with the bond coat is clearly visible. These results confirm the successful deposition of a TBC with luminescence sensing properties throughout the whole thickness.

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Table 1
Crystal lattice parameters of various YSZ:Ln aerogel powders (Ln = Sm, Eu, Dy or Tm) determined by the Rietveld method from XRD spectra.
temperature and thermal gradients sensing capabilities with non-contact phosphor thermometry methods.

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References


Fig. 10. Images of a multilayer TBC prototype (a) under white light illumination, (b) illuminated with a 532 nm laser showing the luminescence intensity at 636 nm from the YSZ:Eu$^{3+}$ layer located at the interface with the bond coat.