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Mid-infrared Raman-soliton continuum pumped by a nanotube-mode-locked sub-picosecond Tm-doped MOPFA

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Abstract: We demonstrate a mid-infrared Raman-soliton continuum extending from 1.9 to 3 µm in a highly germanium-doped silica-clad fiber, pumped by a nanotube mode-locked thulium-doped fiber system, delivering 12 kW sub-picosecond pulses at 1.95 µm. This simple and robust source of light covers a portion of the atmospheric transmission window.

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1. Introduction

Over the past decade, supercontinuum sources in the near-infrared (IR), utilizing silica-based photonic crystal fibers (PCFs), have become a commercial success [1]. Typically, such systems are pumped by a master oscillator power fiber amplifier (MOPFA) [2, 3], employing ytterbium (Yb) doped fiber technology, and can cover the transparency window of silica (~0.35 – 2.2 µm [4, 5]), with high-average spectral power [2, 3, 6].

There is widespread interest in extending the long wavelength edge of the supercontinuum beyond the region where pure silica can be employed, for applications such as spectroscopy of trace gases [7], chemical kinetics [8] and military counter measures [9]. Progress in this regard has been made using fluoride [10, 11] and chalcogenide [12] glass fibers, in particular ZBLAN (ZrF$_4$,BaF$_2$,LaF$_3$,AlF$_3$,NaF) [10, 11, 13] and tellurite [14, 15] glasses, because of their enhanced transmission in the mid-IR [5, 10–16]. Step-index fibers fabricated from soft glasses, with transmission windows extending up to 5 µm, have been used to demonstrate supercontinua covering the 1 – 4 µm region [10, 11], pumped by high-power multi-stage erbium (Er) doped fiber lasers operating near the low anomalous dispersion region of the fiber.
An alternative approach targets the development of PCFs made from tellurite [14, 15], including short length tapers [5], because of their high nonlinear coefficients and broad mid-IR transparency [5, 14, 15].

While fluoride glass fibers have superior transmission beyond 2 µm compared to silica [10, 11], they are non-resistant to moisture [17] and, as such, degrade in air over time [5, 17]. In addition, poor compatibility with silica [5] reduces their potential for realizing fully fiber-integrated systems. Although chalcogenides are hydrophobic [18], consequently more stable against corrosion [5], the fabrication of compound glass fibers exhibiting single-mode performance is complex [5].

Germanium dioxide (GeO₂) is closely related chemically to silicon dioxide, sharing many properties that make it an excellent material for the manufacture of single-mode fibers [19], but with superior mid-IR transmission [19–21]. In addition, bulk GeO₂ glass has nearly an order of magnitude larger Raman scattering cross-section compared to bulk SiO₂ [19, 20] and a higher Kerr nonlinearity [19]. Consequently, GeO₂ doping of the core region is routinely used to enhance the Raman gain of conventional silica-based fibers [19–23]. Advances in fabrication techniques permit GeO₂ doping concentrations in excess of 50 mol. % [19], despite a mismatch in the thermal expansion coefficients of SiO₂ and GeO₂ [19, 20].

Recently, Raman-soliton continuum generation was demonstrated in a highly-doped GeO₂ silica-clad fiber, using an Er-based pump source [24, 25]. The bandwidth of the continuum exceeded 1000 nm, with the long-wavelength limit at 2.5 µm [24, 25]. Extension of the infrared edge of the continuum can be achieved using a longer wavelength pump source [26, 27]. Examples of suitable pump systems include: broadly tunable, ultrashort pulse optical parametric amplifiers (OPAs) and optical parametric oscillators (OPOs) [25–28] and, more recently [13], thulium-(Tm) doped fiber-based systems [13]. OPAs and OPOs, however, are complex and expensive systems and cannot deliver the benefit of a compact design, efficient heat dissipation, and alignment-free operation offered by fiber lasers [5].

Ultrafast sources based on Tm-doped fibers operating around 2 µm are becoming increasingly important to address demands for mid-IR sources [5]. Tm-based mode-locked oscillators have previously been reported employing nonlinear polarization rotation (NPR) and semiconductor saturable absorber mirrors (SESAMs) [5]. NPR and SESAMs, however, can suffer from environmental sensitivity or require complex fabrication [5]. Carbon nanotubes (CNTs) [28–33] and graphene [29, 31, 34–37], have emerged as alternative saturable absorbers (SA) with ultrafast recovery time [38–40], able to support short pulses, and with a number of favorable properties, such as broadband operation [34, 38], and ease of fabrication [31, 35, 41] and integration [30] into all-fiber configurations. While broadband operation is an intrinsic property of graphene [35], in CNTs this can be achieved using a distribution of tube diameters [34]. A variety of techniques have been implemented in order to integrate CNTs and graphene into lasers [29]. CNTs and graphene embedded in polymer matrices can be easily integrated into a range of photonic systems [31, 35]. CNTs can be homogeneously embedded into polymer matrices, resulting in high quality composites [31, 42], exhibiting large modulation depths [30, 33, 42, 43], preferred for fiber lasers [5].

Here, we report the generation of a Raman-soliton continuum, extending beyond 2.5 µm, pumped at 1.95 µm with 12 kW peak power pulses delivered from a nanotube-mode-locked Tm-based MOPFA, in an optimized 3.4 m length of 75 mol. % GeO₂ fiber. We use CNTs with diameter ~1.7 nm, in order to achieve a strong absorption in the 2 µm range [44, 45]. The CNTs are embedded in a polymer matrix, thus forming our SA to mode-lock the seed laser of the MOPFA. This approach provides a robust, long-term stable source of radiation in an important band, coincident with a portion of the atmospheric transmission window.

2. Seed laser

The schematic of the seed oscillator is shown in Fig. 1. This consists of all-fiber integrated components, in order to have an environmentally stable and compact system. A Tm–doped fiber amplifier, with integrated optical isolator (ISO), provides a peak small signal gain of ~25 dB at 1.94 µm, with gain available over a broad bandwidth (full width at half maximum,
FWHM ~60 nm), suitable to support the generation of short pulses. A fiber pigtailed air-gap (20% insertion loss) is used to include an intra-cavity band-pass filter (BPF) for pulse stabilization, with 80% maximum transmission and 11 nm bandwidth, centered at ~1.94 µm. The output signal is delivered through a 30:70 fiber coupler. A polarization controller (PC) allows continuous adjustment of the intra-cavity polarization state.

![Diagram of the mode-locked oscillator](image)

Fig. 1. Schematic of the mode-locked oscillator. TDFA: thulium-doped fiber amplifier, ISO: isolator, BPF: bandpass filter, OC: fiber output coupler, CNT SA: carbon nanotube saturable absorber, PC: polarization controller.

The SA is designed to have absorption coincident with the operating wavelength of the oscillator, centered at 1.94 µm. We use CNTs produced by Catalytic Chemical Vapor Deposition (CCVD) of CH\textsubscript{4} over Mg\textsubscript{1-x}Co\textsubscript{x}O solid solution containing Mo oxide [46]. The catalyst and byproducts are dissolved by treatment with concentrated aqueous HCl solution [46]. The remaining carbon-encapsulated catalytic nanoparticles are removed by air oxidation at 450°C for 1 h in an open furnace, followed by HCl washing to dissolve metal oxides formed during the oxidation step [47, 48]. In order to further purify the sample, oxidation in air at higher temperature (570°C) is carried out for a shorter time (30 min) [48]. The residual material is further washed with HCl to dissolve the remaining metal oxides [47, 48].

The absorption spectrum of the purified CNTs is shown in Fig. 2(a) (black line) with a peak between 1.75 and 2.15 µm. The desired operating wavelength is also highlighted (red dashes). To further investigate their optical properties, we also measure the Raman spectra of the purified CNTs powder at 457 (2.71 eV), 514.5 (2.41 eV), and 632.8 nm (1.96 eV). In the low frequency region, the Radial Breathing Modes (RBMs) are observed [48]. Their energy is inversely related to CNT diameter, \(d_t\) [50–52] by the relation \(\omega_{RBM} = C_1 / d_t + C_2\). We use \(C_1 = 214.4\) cm\(^{-1}\) and \(C_2 = 18.7\) cm\(^{-1}\) from Ref [51]. These were derived in Ref [51] by plotting the resonance energy as a function of inverse RBM frequency without any additional assumptions. However, we also validated our results by using the parameters proposed in Refs [50, 52]. For all excitation wavelengths and samples analyzed, we found a maximum discrepancy of 0.03 nm on the tube diameter. The RBM spectra of the powders [Fig. 2(c)] (black lines) show a broad diameter distribution, spanning the 140–377 cm\(^{-1}\) range. This corresponds to CNTs with ~0.6-1.8 nm diameter. Figure 2(d) plots the Raman spectra in the \(G\) region of purified CNTs (black curves). A weak \(D\) band is observed \([I(D)/I(G) = 0.04]\), indicating small number of defects [53].
A polymer composite is then fabricated via solution processing [30–33, 42]. Purified CNTs are dispersed using a tip sonicator (Branson 540 A, 20 kHz) in water with sodium dodecylbenzene sulfonate (SDBS) as surfactant for 4 h. The dispersion is then ultracentrifuged (Sorvall WX Ultra) at 100,000 g, where g is the gravitational acceleration, for 30 mins. The top 70% dispersion, free from insoluble particles and CNT aggregates, is then decanted. 4 ml are mixed with 120 mg polyvinyl alcohol (PVA) and ultrasonicated again for 30 mins, obtaining a homogeneous and stable dispersion. We use water as the solvent, due to its low boiling point compared to common organic solvents used to disperse CNTs, such as N-Methyl Pyrrolidone (NMP) [31] with a boiling point of 206 °C [54]. SDBS is used as the surfactant for its ability to produce small CNT bundles [55], unlike bile salts, e.g. sodium cholate, more effective in the dispersion of individual nanotubes [55]. PVA is used for its compatibility with water [31]. The CNT-polymer mixture is drop-cast in a petri dish. Slow evaporation, over 4-5 days at room temperature in a desiccator, produces a free-standing ~50 µm composite. The concentration, c, of CNTs in the PVA film is estimated to be ~0.3 weight percent (wt%), derived by measuring the weight of the decanted CNT dispersion compared to that of the solution × 100%.

Figure 2(a) plots the absorption spectra of the PVA (grey line), the CNT-PVA composite (red line) and the pristine CNTs (black line). The absorption of the PVA is ~1 order of magnitude lower with respect to the CNT-PVA composite, in the 400 – 2000 nm range, thus negligible. The absorption band between 1.75 and 2.15 µm corresponds to eh11 excitonic transitions of tubes with diameters in the 1.5–1.8 nm range [43, 44]. Power-dependent absorption is measured with an optical parametric oscillator (Coherent, Chameleon) delivering ~260 fs pulses with 80 MHz repetition rate at 1945 nm. The optical transmittance is determined by monitoring the input and output power on the CNT composite. The
nonlinear transmittance increases from ~76% to ~85%, Fig. 2(b), giving a ~9% change in transmittance, comparable to that typically reported for CNTs [32, 41, 42]. We also measured the Raman spectra of the composite and, for the three excitation wavelengths, we do not observe any change in the RBM distribution with respect to the pristine CNTs. Thus, the dispersion process and the CNT-PVA composite fabrication do not induce additional defects [52] with respect to the starting material.

![Graph](attachment:image.png)

Fig. 3. Seed laser performance. (a) Autocorrelation of the output pulses, with a deconvolved duration of 3.7 ps. (b) The corresponding optical spectrum (high resolution, inset).

The CNT-SA is then inserted between two SMF-28 fiber connectors and directly fusion spliced into the laser cavity. The oscillator operates at the fundamental repetition frequency of the cavity, 6.1 MHz, and produces 3.5 mW average output power, corresponding to 0.57 nJ single pulse energy. The autocorrelation of the output pulses and the corresponding optical spectrum are plotted in Fig. 3. The autocorrelation is well fitted by a \( \text{sech}^2 \) pulse-shape, with a 3.7 ps deconvolved FWHM duration, Fig. 3(a). The optical spectrum, shown in Fig. 3(b) and recorded using an automated grating spectrometer, is centered at 1.94 µm with a 3.2 nm FWHM. The seed spectrum is also recorded using a long wavelength range optical spectrum analyzer (Yokogawa). This allows us to observe solitonic spectral sidebands (Fig. 3(b) inset).

The overall cavity group velocity dispersion (GVD) can be estimated from the spectral sideband separation [56] \( \Delta \lambda = \pm \frac{\lambda^2}{2 \pi c} \frac{\tau}{1.763} -1 \frac{8 \pi \beta_2}{z_c} \), where \( \tau \) [s] is the pulse duration, \( \lambda \) [m] is the peak wavelength, \( z_0 = \frac{\pi \tau^2}{2 \beta_2} \) [s-m] is the soliton period with \( \beta_2 \) [s/m] the GVD coefficient and \( z_c \) [m] the cavity length. Thus, \( \beta_2 = \frac{\pi \tau^2}{2 z_0} = -69.2 \text{ ps}^2 \text{ km}^{-1} \) is estimated.
3. Tm-doped MOPFA and supercontinuum generation

The configuration of the system for supercontinuum generation is shown in Fig. 4. Prior to amplification, the output pulses from the seed oscillator are temporally stretched to greater than 80 ps through dispersive broadening in a 1.2 km single-mode silica fiber, with normal GVD~34 ps² km⁻¹ at 1.95 µm, in order to reduce the peak power.

The amplifier is constructed from 5.5 m single-mode Tm-doped fiber (Nufern SM-TSF-9/125), pumped through a fiber wavelength division multiplexer (WDM) by a 7 W continuous-wave (CW) Er laser. The lengths of active fiber and pump power are optimized to preserve pulse quality during amplification, so to maximize the signal gain. A second WDM at the output of the amplifier extracts residual pump light. Figure 5(a) shows the autocorrelation trace of the amplified output, under maximum pump power, corresponding to an average signal power of 150 mW. Again, a sech² fit well represents the pulse-shape, with ~80 ps duration. The pulse spectrum [Fig. 5(b)] is centered at 1.95 µm and has a 6.2 nm FWHM. A pump power of 7 W is required because the seed line is not coincident with the peak of the spontaneous fluorescence spectrum of the active fiber used in the amplifier, limiting the available gain at 1.95 µm.

Fig. 4. Schematic of the Tm-fiber system for pulse amplification, compression and supercontinuum generation. WDM: wavelength division multiplexer. Mirror M₁ is tilted to separate the outgoing beam from the incoming beam. Mirror M₂ reflects the compressed pulses output without introducing any additional losses. M₁, M₂, and M₃ are highly reflective broadband mirrors.

To achieve a kW level peak power, the pulses are recompressed by collimating the output through an aspheric lens and double passing a pair of 800 lines/mm gold coated gratings, optimized to operate in the 1.8 – 2.2 µm range. Through adjustment of the grating separation, 850 fs FWHM duration is achieved [Fig. 6(a)]. No pedestal component is observed, indicating high quality amplification and compression. The low-level satellite pulses are assigned to residual nonlinear chirp accumulated in the long-length stretcher stage, to the amplification process, and to uncompensated third order dispersion. Figure 6(b) shows the largely unchanged spectrum after compression. 10 cm of SMF is used as an intermediate fiber, to facilitate improved coupling to the GeO₂ fiber. Repeatable splice loss between the GeO₂ and SMF fiber is as low as 0.4 dB (i.e. ~8.8%). The zero dispersion wavelength (ZDW) of the GeO₂ fiber (i.e. the wavelength where the group delay dispersion of a fiber is zero [5]) shifts with the doping concentration, such that at a 75 mol. % this is expected to be in the 1.8 – 1.9
\( \text{µm range [21]. Thus, here we pump in the region of low anomalous GVD and expect continuum dynamics initiated by modulation instability (MI), given the pump format. This explanation of the dynamics [57] is consistent with the fact that the spectral sideband signature of MI is observed at low pump powers, before significant continuum formation.} \)

![Graph](image)

**Fig. 5. MOPFA results.** (a) Autocorrelation of the amplified pulses with 81 ps duration. (b) Corresponding optical spectrum.

Spectral measurements after the GeO\(_2\) fiber are taken using an automated Spex 500 spectrometer in combination with a PbS IR detector and lock-in amplifier. Although the transmission of GeO\(_2\) is superior to silica beyond \( \sim \text{2.1 µm} \), the loss at wavelengths longer than 2.5 µm still increases significantly to hundreds of dB km\(^{-1}\) [20]. As such, it is important to pump short lengths, with high-peak power pulsed sources. Note that this limits the application of CW lasers as a suitable pump source in this case, thus high-average spectral power continuum sources in this wavelength range remain a challenge.

The average spectrum is recorded as a function of fiber length using a cutback technique [58] starting from a 5.6 m GeO\(_2\) fiber. A plot of the corresponding generated supercontinuum bandwidth (10 dB level) as a function of fiber length is shown in Fig. 7. The continuum width monotonically decreases beyond \( \sim \text{3.5 m} \), which can be attributed to re-absorptive loss due to infrared absorption [20]. The broadest spanning supercontinuum achieved, where a balance of nonlinear gain and linear loss is reached, is shown in Fig. 8. A spectrum extending from 1.9 to 3 µm is generated in an optimized fiber length of 3.4 m.
Fig. 6. (a) Autocorrelation of the compressed pulses, with an 850 fs duration. (b) Corresponding optical spectrum.

Fig. 7. Supercontinuum bandwidth (10 dB) as a function of GeO₂ fiber length.
4. Conclusions

We demonstrated the generation of a Raman-soliton continuum, extending from 1.9 to 3 µm in an optimized 3.4 m length of 75 mol. % GeO₂ fiber, and pumped at 1.95 µm by a Tm-based MOPFA delivering 12 kW peak power, sub-picosecond pulses. This robust and simple fiber system addresses an important region beyond the long wavelength extent of common pure-silica PCF-based supercontinuum light sources.

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